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Direct Dioxygen Radical Coupling Driven by Octahedral Ruthenium—Oxygen—Cobalt Collaborative Coordination for Acidic Oxygen Evolution Reaction

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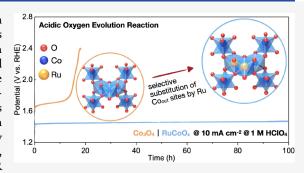
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ABSTRACT: The acidic oxygen evolution reaction (OER) has long been the bottleneck of proton exchange membrane water electrolyzers given its harsh oxidative and corrosive environments. Herein, we suggest an effective strategy to greatly enhance both the acidic OER activity and stability of Co₃O₄ spinel by atomic Ru selective substitution on the octahedral Co sites. The resulting highly symmetrical octahedral Ru–O–Co collaborative coordination with strong electron coupling effect enables the direct dioxygen radical coupling OER pathway. Indeed, both experiments and theoretical calculations reveal a thermodynamically breakthrough heterogeneous diatomic oxygen mechanism. Additionally, the active Ru–O–Co units are well-maintained upon the acidic OER



thanks to the electron transfer from surrounding electron-enriched tetrahedral Co atoms *via* bridging oxygen bonds that suppresses the overoxidation and thus dissolution of active Ru and Co species. Consequently, the prepared catalyst, even with a low Ru mass loading of *ca.* 42.8 μ g cm⁻², exhibits an attractive acidic OER performance with a low overpotential of 200 mV and a low potential decay rate of 0.45 mV h⁻¹ at 10 mA cm⁻². Our work suggests an effective strategy to significantly enhance both the acidic OER activity and stability of low-cost electrocatalysts.

■ INTRODUCTION

Water electrolysis is considered as an environmental-friendly technology for green hydrogen production. Compared with the commercially available alkaline water electrolyzer, the acidic proton exchange membrane water electrolyzer (PEMWE) possesses several advantages, including lower ohmic resistance, higher current densities, better compatibility with sustainable energy resources, etc.¹⁻³ The development of the PEMWE, however, has long been limited by the anodic oxygen evolution reaction (OER), due to the lack of effective and stable electrocatalysts as most known electrocatalysts would suffer from corrosion and dissolution under the strongly acidic and oxidative conditions. Even for the state-of-the-art noble metal Ru-based materials, the stability is still unsatisfactory because of the formation of soluble species (e.g., RuO₄) during the acidic OER.4-6 Previous studies also demonstrated that the OER over RuO₂ is dominated by lattice oxygen mechanism (LOM), which involves the continuous insertion and extraction of lattice oxygens and thus would destroy the structural stability.7-10

In addition to the severe performance decay, the high price and scarcity of Ru (ca. \$17,420 kg⁻¹) are also a big concern. The noble metal content in Ru-based catalysts (e.g., transition metal-doped RuO₂, ^{11–13} RuB₂, ¹⁴ Ru/RuS₂¹⁵) is usually more than 80 wt %, and the Ru mass loading is typically higher than

0.2 mg cm⁻² in three-electrode tests or 2 mg cm⁻² in PEMWE in order to achieve decent performance. Therefore, enormous efforts have been devoted to lowering the Ru usage by improving the utilization efficiency (e.g., through loading highly dispersed clusters or even single atoms on acid-resistant supports) or further, by directly using non-noble catalysts. 16-20 In this regard, Co₃O₄ has been considered as a promising alternative. Previous studies have revealed that Co₃O₄ delivers an acceptable OER stability in acidic electrolytes, yet its activity requires further enhancement.^{21–24} The Co₃O₄ shares a spinel AB₂O₄ structure, where Co^{II} ions occupy the tetrahedral A sites and Co^{III} ions occupy the octahedral B sites, respectively. The latter has been demonstrated to be significantly more active than the former for OER. 25 This is confirmed by the fact that the ZnCo2O4 (where Zn atoms occupy the tetrahedral sites only) shows comparable OER activity as that of Co₃O₄, ²⁶ while the CoAl₂O₄ (where Al atoms

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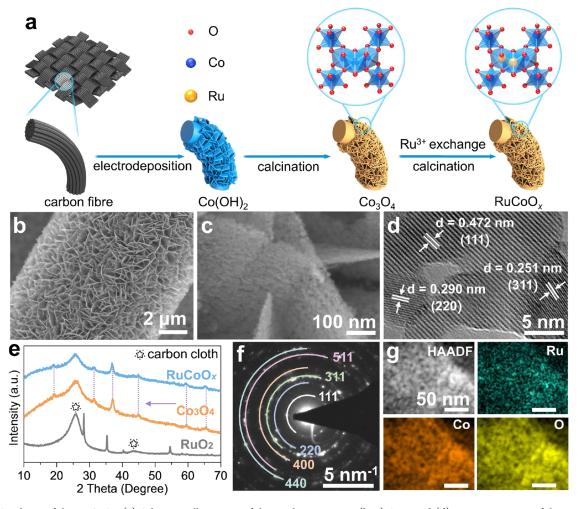


Figure 1. Synthesis of the RuCoO_x. (a) Schematic illustration of the synthetic process. (b, c) SEM and (d) HRTEM images of the RuCoO_x. (e) XRD patterns of the RuO₂, Co₃O₄ and RuCoO_x. (f) SAED patterns of the RuCoO_x. (g) HAADF-STEM image and the corresponding elemental maps of the RuCoO_x.

occupy the octahedral sites) possesses weaker activity than Co_3O_4 . The engineering of octahedral Co (*i.e.*, Co_{oct}) sites, for instance, by atomic substitution, therefore provides an effective way to regulate the coordination environment of Co_3O_4 and further boost the OER performance. In addition, previous studied have demonstrated a diatomic oxygen mechanism (DOM) over Co_3O_4 for OER. $^{28-32}$ The DOM proceeds with direct dioxygen radicals coupling to release O_2 , which can effectively circumvent the formation of the *OOH intermediate and thus break the *OH and *OOH scaling relationship limitation of the traditional adsorbate evolution mechanism (AEM). 33,34 In this regard, the precise substitution of octahedral Co with other atoms might also enable the DOM pathway and significantly enhance the acidic OER performance, though it has not been realized yet.

In this work, we developed a modified Co_3O_4 spinel electrocatalyst with atomic Ru substitution on the Co_{oct} sites (named as $RuCoO_x$ thereafter). The construction of highly symmetrical $Ru_{oct}-O-Co_{oct}$ collaborative coordination with a strong electron coupling effect significantly shortens the spatial distance between Ru_{cot} and Co_{cot} which makes the direct dioxygen radical coupling become favorable. Indeed, we confirmed both experimentally and theoretically that a thermodynamically breakthrough heterogeneous DOM driven by the $Ru_{oct}-O-Co_{oct}$ collaborative coordination is preferred,

rather than the theoretically limited AEM and the structurally unstable LOM that are more commonly observed. More importantly, the strong interaction of the Ru_{oct} –O– Co_{oct} collaborative coordination is also well-maintained upon the acidic OER due to the electron capture from surrounding electron-enriched tetrahedral Co (*i.e.*, Co_{tet}) atoms *via* bridging oxygen bonds. As a result, the $RuCoO_x$ exhibits significantly enhanced OER activity and stability in acid even with a low Ru loading of *ca.* 42.8 μg cm⁻². Specifically, the $RuCoO_x$ delivers a small overpotential of 200 mV at 10 mA cm⁻² and can steadily operate for at least 100 h with an only 45 mV potential decay. Further, the assembled PEMWE based on the commercial 20 wt % Pt/C and $RuCoO_x$ combination also shows outstanding stability even at a higher current density of 100 mA cm⁻².

■ RESULTS AND DISCUSSION

Catalyst Synthesis and Structural Characterization. The $RuCoO_x$ catalyst with Ru_{oct} substitution was synthesized on carbon cloth substrate by a cation exchange strategy. In a typical synthesis, the $Co(OH)_2$ nanosheets were first electrodeposited onto carbon cloth followed by calcination to produce the porous Co_3O_4 nanosheets precursor, which was then immersed in a $RuCl_3$ aqueous solution to initiate the cation exchange. The resulting product was then annealed into the anhydrous $RuCoO_x$ nanosheet catalyst (Figure 1a, see the

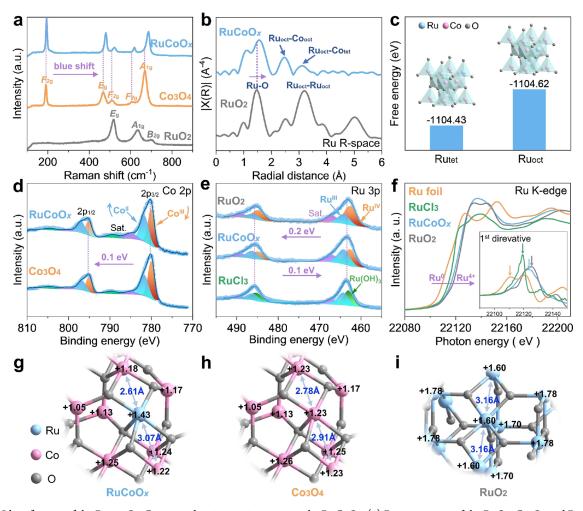


Figure 2. Identification of the Ru_{oct} –O– Co_{oct} coordination environment in the $RuCoO_x$. (a) Raman spectra of the RuO_2 , Co_3O_4 and $RuCoO_x$. (b) Ru FT-EXAFS spectra of the RuO_2 and $RuCoO_x$. (c) Energy diagram of a substituted Ru atom in the Co_3O_4 . (d) Co 2p XPS spectra of the Co_3O_4 and CoO_x . (e) Ru 3p XPS spectra of the RuO_2 , $RuCoO_x$, and $RuCoO_x$. (e) Ru 3p XPS spectra of the RuO_2 , $RuCoO_x$, and $RuCoO_x$, and $RuCoO_x$, inset is the Ru K-edge 1st derivative spectra. Bader charge distribution diagrams of (g) $RuCoO_x$, (h) Co_3O_4 , and (i) RuO_2 .

principle of the cation exchange strategy in the Supporting Information). We systematically investigated the impacts of synthetic conditions and found that a cation exchange time of 9 h and a second calcination temperature of 250 °C would result in the optimal OER activity and operation stability (Figure S1). The conditions were then used for the rest of the study.

We first examined the structural difference between the Co₃O₄ and RuCoO_r. The SEM images show that both the porous Co₃O₄ and RuCoO₂ nanosheet array uniformly covers the surface of carbon fibers (Figure 1b,c and Figure S2a-c), forming an interconnected 3D network with close surface areas (12.36 and 13.36 m² g⁻¹, respectively, Figure S3a). A careful observation further reveals that both the Co₃O₄ and RuCoO_x nanosheets are composed of abundant micropores with the same pore diameter distribution from 5 to 60 nm (Figures S2d, S3b, and S4). Although the RuCoO_x largely inherits the overall morphology from the Co₃O₄ precursor, the lattice distance is sensitively expanded (e.g., $d_{(311)}$ from 0.244 to 0.251 nm, and $d_{(220)}$ from 0.285 to 0.290 nm) as revealed by the highresolution transmission electron microscopy (HRTEM) image (see comparison in Figure 1d and Figure S2e). This can be also confirmed by the X-ray diffraction (XRD) patterns (Figure 1e and Figure S5). Besides the two broad diffraction peaks

originated from the carbon cloth substrate (Figure S6), all the other diffraction peaks can be assigned to spinel Co₃O₄ (JCPDS card no. 74-1656). Notably, the overall diffraction peaks of the RuCoOx slightly shift to lower angles in comparison to those of the Co₃O₄, implying that the lattices of the Co₃O₄ are expanded because of the incorporation of Ru atoms with a larger radius. Moreover, no other Ru species are observed even after the thermal calcination of the RuCoO_x at 800 °C in an Ar atmosphere (Figure S5), which not only rules out the existence of amorphous Ru species but also demonstrates the high structural stability of Ru atoms in the Co₃O₄ matrix. Indeed, the selected area electron diffraction (SAED) pattern of the RuCoO_x also confirms that all the visible polycrystalline rings belong to Co₃O₄ (Figure 1f). Additionally, the high-angle annular dark-field scanning TEM (HAADF-STEM) image and its corresponding elemental maps reveal that the elemental distribution of the substituted Ru is highly consistent with that of Co and O signals (Figure 1g). These results suggest that the Ru atoms were doped into the Co₃O₄ lattice, instead of aggregating on the surface of the Co₃O₄ nanosheets as Ru or RuO₂ clusters.

The mass loadings of Ru and Co on carbon cloth were *ca.* 0.0428 and 2.3495 mg cm⁻², respectively, corresponding to a bulk Ru ratio of *ca.* 1.55 at. % (determined by inductively

coupled plasma mass spectrometry or ICP-MS, see details in the Supporting Information). Meanwhile, the surface Ru ratio, determined by X-ray photoelectron spectroscopy (XPS) analysis, was up to ca.~25 at. % (Figure S7). These results suggest that cation exchange reaction would lead to a concentration gradient of Ru from high to low along the surface to bulk, which improves the Ru utilization efficiency for the surface electrocatalysis reaction. The commercial RuO₂ on carbon cloth with a mass loading of 1 mg cm⁻² was also prepared for comparison (see detailed characterizations in Figure S8).

Identification of Selective Ru Sites in the RuCoO_x. In order to reveal the selective substitution of Co sites (octahedral or tetrahedral) in the Co₃O₄ by Ru atoms, we then conducted the Raman test, which is a sensitive technology to reveal the changes in the coordination environment. The Co_3O_4 spinel presents five Raman-active modes (i.e., A_{1g} , E_g) and three F_{2g} modes, Figure 2a). The characteristic Raman peaks of the RuCoO_x are almost identical to those of Co₃O₄ but shift to the higher frequency region. Notably, the shift of A_{1g} (685 cm⁻¹) is more significant than that of F_{2g} band (195 cm⁻¹). Since the former is related to the stretching of Co^{III}–O in the octahedral (CoO₆) units while the latter raises from Co^{II}-O in the tetrahedral (CoO₄) groups, ^{35,36} thus we can reasonably infer that the Ru atoms preferentially occupy the position of Co_{oct} instead of Co_{tet} sites. Indeed, the decrease in peak intensity of A_{1g}/F_{2g} bands of the RuCoO_x compared to that of the Co₃O₄ also suggests the decrease in the ratio of Co_{oct}/Co_{tet} due to the Ru_{oct} substitution.

We also employed the Fourier transformed (FT) k^3 weighted extended X-ray absorption fine structure (FT-EXAFS) analysis to identify the bonding environment. The Ru FT-EXAFS spectra indicate that a characteristic peak at 2.49 Å observed for the RuCoO_x in the second shell (slightly larger than the Co_{oct}-Co_{oct} bond, 2.45 Å, without phase correction, Figure S9b) can be assigned to Ru_{oct}-Co_{oct} coordination, suggesting that Ru atoms successfully substitute partial Co_{oct} sites in the Co₃O₄. It is noted that the weakened and broadened FT peak at 3.13 Å in the higher shell likely belongs to the Ru_{oct}-Co_{tet} coordination, which is obviously different from the intense FT peak at 3.19 Å that assigned to the Ru_{oct}-Ru_{oct} coordination in the RuO₂. This can also exclude the existence of the pure RuO₂ phase in the RuCoO_x well in line with the SAED and XRD results. Apart from the experimental spectroscopy characterizations, density functional theory (DFT) calculation also reveals that the required free energy of substituting one Ru atom into the Co_{oct} site is 0.19 eV lower than that into the Cotet site (Figure 2c), further confirming the selective occupancy of Ru atoms in the octahedral Co sites of Co₃O₄.

Revelation of Strong Electron Coupling Effect in the RuCoO_x. The Ru_{oct} substitution in the Co₃O₄ is expected to modulate the surface electronic distribution and bonding environment. The surface-sensitive Co 2p XPS spectrum of the RuCoO_x shows a upshift of *ca.* 0.1 eV compared to that of the Co₃O₄ (Figure 2d), suggesting the decrease in the Co_{oct}/Co_{tet} ratio (Table S1). This is not only ascribed to the occupied octahedral sites by substituted Ru atoms as revealed before but also attributed to the partial electron transfer from Ru_{oct} to Co_{oct} atoms *via* bridging oxygen atoms in the RuCoO_x. The later can be further verified by the complementary Ru 3p XPS spectra (Figure 2e), wherein the overall binding energy of the RuCoO_x is 0.2 eV higher than the RuO₂ but 0.1 eV lower than

that of the RuCl₃ control, and a more obvious trend was observed for the Ru 3d XPS spectra (Figure S10), meaning that the Ru oxidation state in the RuCoO_x is between +3 and +4 (Table S1), 39,40 higher than the original Co_{oct} site (+3) in the Co₃O₄. It is noted that the existence of Ru(OH)₃ is inevitable due to the slight hydrolysis of the RuCl₃ in air.³⁹ The Ru oxidation state in the RuCoO_x can be further identified by the Ru K-edge XANES and 1st derivative spectra (Figure 2f), wherein the absorption edge of the RuCoO_x is higher than that of the RuCl₃ but lower than that of the RuO₂. Meanwhile, the energy of the maximum of 1st derivative of the RuCoO_x displays the same trend (see detailed explanation in the Supporting Information). The lower Ru oxidation state in the RuCoO_x than that in the RuO₂ would thus result in the longer bond length of Ru–O bonds (1.56 vs 1.47 Å, Figure 2b), which could be attributed to the increased extranuclear electron numbers of Ru centers in the RuCoO_x. The increased electron cloud density around active Ru sites has been demonstrated to be favorable to facilitating the acidic OER progress. 12,13,41 It is noted that the Co K-edge XANES spectra indicate that the Co oxidation state in the RuCoO_x is almost identical to that in the Co₃O₄ (Figure S9a), and the corresponding distances of Co-O, Cooct-Cooct and Cooct/tet-Cotet bonds in R space of FT-EXAFS spectra are also almost identical (Figure S9b). This is not surprising because the X-ray absorption spectroscopy is bulk sensitive, while the ion exchange most likely occurs on the surface of the Co₃O₄ given the trace amount of Ru (1.55 at. %) and thus would not significantly affect the bulk properties.

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Considering that the XPS and XANES only identify the overall change of electronic structure of the Co₃O₄ after Ru_{oct} substitution, we further conducted the Bader charge analysis to explore the partial change of elemental charge distribution on the catalyst (near-)surface in detail. The substituted Ru_{oct} atom can significantly modulate the positive charge of one of the oxygen-bridging Co_{oct} atoms from +1.23lel to +1.18lel and slightly change the charge distribution of the other neighboring Co_{oct} sites, yet barely affect the surrounding Co_{tet} sites (see comparison between the $RuCoO_x$ and Co_3O_4 in Figure 2g,h). In addition, the Bader charge of the substituted Rucot atom (+1.43lel) in the RuCoO_x is obviously higher than that of the original Co_{oct} atom (+1.23lel) in the Co₃O₄ but still lower than that of Ru atoms in the RuO₂ (e.g., +1.78, +1.70, and +1.60lel, Figure 2i), which also strongly supports the Ru 3p XPS and Ru K-edge XANES results. Both results reveal that Ruoct substitution in the Co₃O₄ would induce electron transfer from the Ruoct atom to an adjacent Cooct atom via bridging oxygen bonds, well in line with the XPS analysis. The resulted partial charge imbalance in the RuCoO_x remarkably enhances the hybrid behavior of Ru_{oct}-O and Co_{oct}-O coordination, resulting in the strong electron coupling effect of the Ru_{oct}-O-Co_{oct} unit with the shortened atomic distance between Ru_{oct} and Co_{cot} (2.61 Å) in comparison to the corresponding Co_{oct}-Co_{oct} atomic distance (2.78 Å) in the Co₃O₄ (Figure 2g,h). The heterogeneous metal sites with shortened distance in the Ru_{oct}-O-Co_{oct} collaborative coordination (research focus thereafter) is more beneficial than neighboring Ru_{oct}-Co_{oct} sites (3.07 Å) to conducting direct dioxygen radical coupling for the heterogeneous DOM. It is noted that the various Bader charges of Co in the Co₃O₄ and RuCoO_x can be attributed to the different coordination environments of (near-)surface and bulk atoms (Figure S11).

Apart from the local Bader charge analysis, we also calculated the partial band centers and adsorption ability of

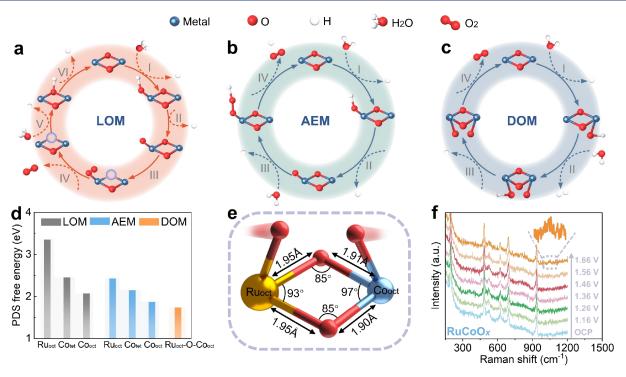


Figure 3. Identification of OER mechanism of the RuCoO_x. Schematic illustration of (a) LOM, (b) AEM, and (c) DOM reaction pathways. (d) Comparison of the PDS energy barrier of three reaction pathways. (e) Simulative structural model of a Ru_{oct}-O-Co_{oct} unit with two adsorbed oxygen radicals. (f) *In situ* Raman spectra under applied potentials.

water molecules. The result indicates that the substituted Ru_{oct} atom in the Co₃O₄ can adjust the 3d band center of adjacent Co_{oct} sites (from -1.41 to -1.48 eV) more significantly than that of surrounding Co_{tet} sites (from -1.27 to -1.23 eV, Figure S12a,b). Additionally, the Ruoct 4d band center in the RuCoO_x also distinctly downshifts compared with those in the RuO_2 (-2.00 vs -1.37 or -1.46 eV, Figure S12a,c). Compared with Co sites in the Co₃O₄ and Ru sites in the RuO₂, the optimized band centers in the RuCoO_x contribute to strengthening the water molecule adsorption ability of the Ru_{oct}-O-Co_{oct} units in acid (Figure S13), which is favorable to accelerating the subsequent OER steps. Interestingly, we found that the O 2p band centers in the Ru_{oct}-O-Co_{oct} units also display the obvious downward shift in comparison to those in the Co₃O₄ and RuO₂ (Figure S12), suggesting that lattice oxygens would not participate in the OER pathway, i.e., the LOM pathway is not preferred. Whereas, for the RuO2, its O 2p band centers are closest to the Fermi level (Figure S12c). Meanwhile, the formed strongest Ru 4d-O 2p orbital hybridization (the closest band center distance between Co 3d/Ru 4d and O 2p orbitals, Figure S12) would facilitate the lattice oxygen activation and subsequently trigger the LOM reaction pathway, 8,18,42 thereby causing inevitable structural deformation or even collapse and further the performance decay. Indeed, our DFT calculation reveals that the rutile RuO₂ (110) facet with the lowest thermodynamic surface energy prefers to follow the LOM rather than the AEM reaction pathway (Figure S14, Tables S2 and S3). The LOM on the RuO₂ was also experimentally confirmed by ¹⁸O isotope-labeled differential electrochemical mass spectrometry (DEMS) in previous studies.^{7–10} In addition to the formation of Ru_{oct}-O-Co_{oct} collaborative coordination with a strong electron coupling effect and enhanced water adsorption ability, the substitution of Co_{oct} sites by Ru atoms might also slightly

cause structural distortion and thus create defects. Both the O 1s XPS spectra and electron paramagnetic resonance (EPR) analysis reveal that the $RuCoO_x$ possesses more oxygen vacancies than the Co_3O_4 and RuO_2 (Figures S15 and S16, Table S1).

Identification of Acidic OER Mechanism of the RuCoO_x. Given the unique coordination environment and strong interaction of the \bar{Ru}_{oct} -O-Co $_{oct}$ units in the RuCoO $_{s}$, we then conducted DFT calculation to explore the OER reaction pathway. According to the XRD result and previously reported articles, we used the spinel Co₃O₄ (311) facet with Ru_{oct} substitution as the model (Figure S17).^{22,43} We first evaluated the two well-accepted OER mechanisms (i.e., AEM and LOM) for the $RuCoO_x$ (Figure 3a and b). The calculation result indicates that the potential determining step (PDS) of the RuCoO_x following the AEM reaction pathway is the dehydrogenation step on the Cooct sites (1.88 eV, Figure S18a, Table S2), while the O-O coupling step by lattice oxygen activation follows the LOM reaction pathway at the same sites (2.07 eV, Figure S18b, Table S3). The maximum Gibbs free energy of the AEM is lower than that of the LOM reaction pathway (Figure 3d), meaning that the AEM pathway is preferred over the RuCoO_x. However, we consider that the Ru sites are much more active than Co_{oct} sites in theory, 44,45 and in fact, the OER process over Ru_{oct} sites in the RuCoO_x is only significantly hindered by the transformation of *O to *OOH intermediates (2.43 eV), while the adsorption of *OH intermediate (0.25 eV) and the subsequent dehydrogenation (0.75 eV) proceed much easier (Figure S18a, Table S2). This indicates that the OER over the RuCoO_x could possibly proceed with an unusual DOM reaction pathway driven by the heteroatomic Ru_{oct} -O- Co_{oct} units. The DOM involves the adsorption of the *OH intermediate, the subsequent dehydrogenation, and the final dioxygen radical coupling

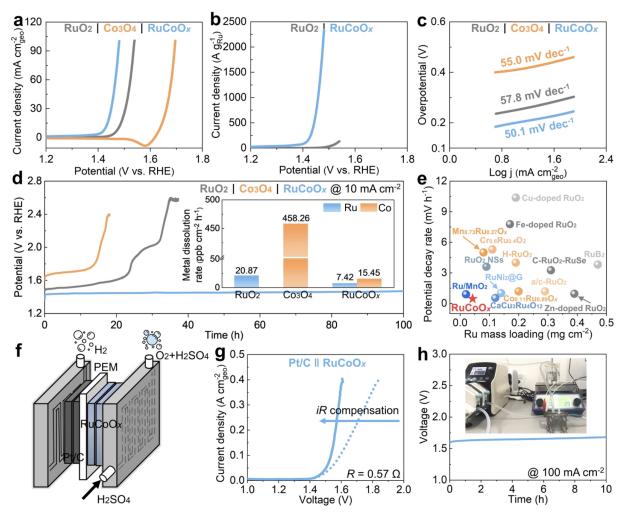


Figure 4. Acidic OER performance of the RuCoO_x. (a) Geometric LSV curves of the RuO₂, Co_3O_4 and RuCoO_x. (b) Ru mass normalized LSV curves of the RuO₂ and RuCoO_x. (c) Tafel plots of the RuO₂, Co_3O_4 and RuCoO_x. (d) Chronopotentiometry curves of the RuO₂, Co_3O_4 , and RuCoO_x at 10 mA cm⁻² (inset shows the average metal dissolution rate). (e) Comprehensive evaluation of potential decay rates and Ru mass loading of the RuCoO_x and other recently reported high-performance acidic OER electrocatalysts. (f) Schematic illustration of the PEMWE configuration using the RuCoO_x and Pt/C couple. (g) Geometric LSV curves of the PEMWE. (h) Chronopotentiometry curves of the PEMWE at 100 mA cm⁻² (inset shows the optical photograph of the device).

process (Figure 3c), which has been widely studied in theoretical and experimental aspects. $^{28-34}$ We then further calculated the bond lengths and the bond angles of the Ru_{oct}-O-Co_{oct} units and found that they are highly symmetrical (Figure 3e) along with shortened atomic distances (Figure 2g). Both are beneficial to conducting the final O-O radical coupling with a low thermodynamic energy barrier. 34,46,47 Indeed, our simulation shows that the required free energy of the dioxygen radical coupling to form O₂ is as low as 1.17 eV (Figure S19a, Table S4), which further suggests that the OER over the RuCoO_r might proceed with the DOM reaction pathway. In this case, the PDS would be the dehydrogenation of *OH intermediate on the Ru_{oct} site (step III in Figure 3c), which requires a free energy of 1.74 eV (Figure 19a, Table S4), lower than that of the RuCoO_xvia the AEM or LOM reaction pathway (1.88 and 2.07 eV, respectively, Figure 3d) and that of the RuO₂ following the favorable LOM reaction pathway (2.16) eV, Figure S14). In addition, we also investigated the Ru_{oct}-O-Co_{tet} coordination environment, and the bond lengths of Ru_{oct}-O and Co_{tet}-O are 2.00 and 1.87 Å, respectively (Figure S20). The unsymmetrical Ru_{oct}-O-Co_{tet} units would lead to a much higher energy barrier for the dioxygen radical

coupling (2.11 eV, Figure S19b, Table S4), which thus becomes the PDS. We also evaluated dual Ru_{oct} sites, dual Co_{oct} sites, and the combination of Co_{oct} and Co_{tet} sites in the $RuCoO_x$ (Figure S19c–e, Table S4). The result demonstrates that the thermodynamic free energies of these dual sites at the PDSs are higher than that of the cooperative Co_{oct} and Ru_{oct} sites. Accordingly, the DFT calculation results reveal that the OER over the $RuCoO_x$ catalyst proceeds with the heterogeneous DOM reaction pathway thanks to the highly symmetrical and contractive Ru_{oct} –O– Co_{oct} collaborative coordination, which has not been discovered so far.

The dioxygen radical coupling, which can be captured by *in situ* electrochemical techniques, ^{34,48–50} reliably distinguishes the DOM from the AEM and LOM. To experimentally confirm the O–O radical coupling, we then carried out the *in situ* Raman test in 0.1 M HClO₄. As shown in Figure 3f, when the applied potential is up to 1.66 V *vs* RHE (without *iR* correction) for the RuCoO_x, the Raman shift at around 1050 cm⁻¹ assigned to dioxygen radical coupling appears, ^{50–52} which strongly supports the heterogeneous DOM reaction pathway of the RuCoO_x. We noted that the signal is relatively weak, but unfortunately failed to enhance the intensity either

by prolonging the test time or applying higher potentials, as the large amount of generated oxygen bubbles would interfere with the Raman incident light. It is also noted that the Raman signals assigned to Co–*OOH (~600 cm $^{-1}$) or Ru–*OOH (~720 cm $^{-1}$) are not detected, thereby excluding the possibility of the AEM over the RuCoO $_x$ - 23,50,53 In brief, we confirmed both theoretically and experimentally that the RuCoO $_x$ catalyst prefers a thermodynamically breakthrough heterogeneous DOM driven by the Ru $_{\rm oct}$ –O–Co $_{\rm oct}$ collaborative coordination.

Acidic OER Performance of the RuCoO_x. Given that the DOM can effectively circumvent the thermodynamic energy barrier step (i.e., the formation of the *OOH intermediate) and thus break the *OH and *OOH scaling relationship limitation of the AEM, the RuCoO_r catalyst is expected to exhibit higher activity than the state-of-the-art RuO₂ and Co₃O₄ catalysts. To further verify the inference, we then conducted the acidic OER measurements in a 1 M HClO₄ electrolyte that is much harsher than the commonly used 0.1 M HClO₄. The Hg/Hg₂SO₄ reference electrode was calibrated against reversible hydrogen electrode (RHE) using a cyclic voltammetry (CV) method (Figure S21).54 Linear sweep voltammetry (LSV) curves display that the RuCoO_x only requires a small overpotential of 200 mV to reach the benchmark current density of 10 mA cm⁻², much lower than that of the RuO₂ and Co₃O₄ (265 and 409 mV, respectively, Figure 4a). It is noted that the activity of the commercial RuO₂ is comparable to the reported value in the literature (Table S5). Although the RuCoO_x exhibits a comparable geometric activity to other recently reported high-performance acidic OER electrocatalysts, its mass activity (e.g., 2278 A g_{Ru}⁻¹ at 1.48 V vs RHE, Figure 4b) is significantly higher than the RuO₂ and most of the Ru or Ir-based catalysts (Figure 4b, Table S6). We also measured the electrochemically active surface area (ECSA) by evaluating the double layer capacitance $(C_{\rm dl})$ (Figures S22–S24). The result shows that the calculated ECSA of the RuCoO_x (5.70 cm²) is larger than that of the RuO₂ and Co₃O₄ (1.58 and 5.42 cm⁻², respectively). It is noted that the RuCoO_x still possesses the largest ECSA or BET normalized currents (Figures S25 and S26), suggesting the highest intrinsic activity among the three catalysts. This is further confirmed by the turnover frequency (TOF), which presents the transfer number of molecules per surface active site per unit time. For example, the RuCoO_x achieves an 8.5fold higher TOF (0.51 s^{-1}) than the RuO₂ (0.06 s^{-1}) at 1.48 V vs RHE, while the Co₃O₄ has no detectable OER currents at the same potential (Figures S27 and S28).

Besides the high activity, the RuCoO_x also possesses the smallest geometric Tafel slope among three catalysts (Figure 4c), suggesting that the Ru_{oct}–O–Co_{oct} coordination can accelerate the acidic OER kinetics. This is further confirmed by the electrochemical impedance spectroscopy (EIS) tests at a fixed potential of 1.43 V ν s RHE, at which the RuCoO_x exhibits the smallest film resistance (R_f) and charge transfer resistance (R_{ct}) compared to the RuO₂ and Co₃O₄ (Figure S29, Table S7). It is noted that we also fit the OER capacitance behavior (*i.e.*, Z_{CPE2}) of three electrodes and found that the trend is almost in accordance with the BET and roughness factor results (Table S8). All the evidence reveals that the thermodynamically breakthrough DOM contributes to the enhanced acidic OER activity of the RuCoO_x, superior to the Co₃O₄ and RuO₂.

We further examined the stability of the three catalysts at the benchmark current density of 10 mA cm⁻² (Figure 4d). The operation stability of the OER electrocatalysts in acids has long been the bottleneck. Most transition metal oxides, including Co₃O₄, ²⁴ WO₃, ⁵⁵ and MnO₂, ⁵⁶ cannot tolerate the harsh corrosive conditions. Even for the current state-of-the-art noble RuO₂ catalysts, the stability is still a big concern as they would easily form acidic soluble RuO₄ species with higher oxidation states under the highly oxidative and acidic environments.^{4,5} Indeed, our result confirms that both the RuO2 and Co3O4 suffer from fast performance decay with quickly elevated potentials within a few hours. Impressively, the RuCoO_x can steadily operate for at least 100 h with an average degradation rate of only 0.45 mV h⁻¹. The acidic OER performance of the RuCoO_x, evaluated by both the low potential decay rate and noble metal mass loading, is superior to various recently reported high-performance Ru-based catalysts. (Figure 4e, and a more detailed comparison in Table S6). Notably, the average metal dissolution rate of the RuCoO_x during the stability test is significantly slower than that of the RuO₂ and Co₃O₄, which is only $\sim 1/3$ and $\sim 1/30$ as that of the latter two, respectively (determined by ICP-MS, inset of Figure 4d). The stability number (S-number), defined as the amount of evolved O2 when a metal atom dissolves in the electrolyte, 57 is another vital metric to evaluate the acidic OER stability and economic benefit. Since the cost of Ru (ca. \$17,420 kg⁻¹) is nearly 400 times as that of Co (ca. \$46 kg⁻¹), we therefore separately calculated the S-number of the RuCoO_x based on dissolved Ru or Co metal. The calculated result indicates that the S-number of the RuCoO_x per dissolved Ru atom (31774) is nearly 3 times as that of the RuO₂ (11297), while the S-number of the RuCoO_x per dissolved Co atom (8895) is nearly 30 times as that of the Co₃O₄ (300) (Figure S30). In addition, the LSV curve of the RuCoO_x after 1000 accelerated CV scans shows negligible changes (Figure S31), suggesting that it can withstand the frequent and intermittent power switching

We finally assembled a proton exchange membrane water electrolyzer (PEMWE) using two pieces of the RuCoO_r as the anode and commercial 20 wt % Pt/C as the cathode, respectively (Figure 4f, see details in the Supporting Information). The PEMWE requires cell voltages of only 1.53, 1.56, and 1.60 V (after iR compensation) to drive the large current densities of 100, 200, and 400 mA cm⁻², respectively (Figure 4g). Moreover, the device can steadily operate at 100 mA cm⁻² for continuous 10 h without obvious voltage fluctuation (Figure 4h), and the corresponding Faradaic efficiencies of H₂ and O₂ production reach ca. 97.29 and 90.94%, respectively (determined by water drainage method, Figure S32). The reduced Faradaic efficiency of the OER might be ascribed to the bubble accumulation in the rubber tubes and the possible slight oxidation of the carbon cloth substrate. It is worthy that the mass loading of precious Ru (ca. 85.6 μ g cm⁻²) used in the PEMWE is at least 20-fold less than that of other Ru-based PEMWE. 53,58,59

Origin of the Acidic OER Stability of the RuCoO_x. The structural integrity of the RuCoO_x after the 100 h of chronopotentiometry test was also investigated by the post physical characterizations. Electron microscope images and XRD patterns reveal no significant changes in the overall morphology and crystal structure (Figures S33 and S34). Moreover, the EPR spectra (Figure S35), together with the O 1s XPS spectra (Figure S36, Table S1), suggest an obvious

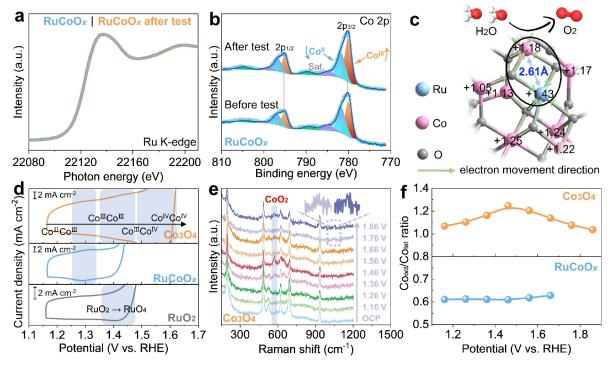


Figure 5. Acidic OER stability analysis. (a) Ru K-edge XANES and (b) Co 2p XPS spectra of the RuCoOx before and after the chronopotentiometry test. (c) Schematic illustration of electron migration in the RuCoO_x. (d) CV curves of the RuO₂, Co₃O₄, and RuCoO_x. (e) In situ Raman spectra of the Co₃O₄. (f) Co_{oct}/Co_{tet} ratio of the Co₃O₄ and RuCoO_x under applied potentials.

decrease in oxygen vacancy concentration. This confirms that the oxygen vacancies are filled upon the OER, since the DOM requires a highly integral and symmetric structure. More importantly, we found that the Ru oxidation state is perfectly maintained after the long-term stability test, as revealed by the Ru K-edge XANES and XPS spectra (Figure 5a, Figure S37, and Table S1), though the Ru_{oct}-Co_{oct} coordination is slightly weakened possibly due to the trace Ru dissolution, as revealed by Raman and FT-EXAFS spectra (Figures S38 and S39). This suggests the high structural integrity of the existing Ru_{oct}-O-Co_{oct} units and thus the high acidic OER stability. In addition, the Ru_{oct}-Co_{tet} coordination is slightly stretched (Figure S39), which is likely due to the electrooxidation of Co_{tet} sites during the acidic OER. Indeed, the Co 2p XPS spectra indicate that the overall Co_{oct}/Co_{tet} ratio on the RuCoO_r surface marginally increases (Figure 5b, Table S1). This is not surprising since the relatively electron-deficient Ru_{oct} -O-Co_{oct} units with the strong electronic coupling effect can drag electrons from the adjacent electron-enriched Co (especially Cotet) sites via bridging oxygen bonds during the OER electrocatalysis (Figure 5c), which effectively prevents the overoxidation of active Ruoct and Cooct sites into acidic soluble RuO4 and CoO2 species, respectively, and thus maintains the acidic OER durability.

The above results confirm the structural integrity of active Ru_{oct}-O-Co_{oct} collaborative coordination, which can also be reflected by comparing the CV curves of the three catalysts (Figure 5d and Figure S40). Three pairs of obvious redox couple at around 1.29, 1.45, and 1.56 V vs RHE for the Co₃O₄ are observed before OER currents, which can be assigned to the Co^{II}Co^{III}/Co^{III}Co^{III}, Co^{III}Co^{III}/Co^{III}Co^{IV}, and Co^{III}Co^{IV}/Co^{IV}Co^{IV} redox reactions, respectively.^{23,61} It is noted that though CoO2 has been regarded as the active OER species in alkaline electrolytes for the majority of Co-based catalysts, 62-64

it is highly dissolvable in acid, and thus the formation of CoO₂ would significantly degrade the OER stability (Figure 4d). For the RuO₂, we also observed a weak reduction peak around 1.41 V vs RHE assigned to RuO₄/RuO₂ redox reaction that is detrimental to the acidic OER durability.⁶⁵ Nonetheless, we only observed a pair of reversible and weak redox couple corresponding to the Co^{II}Co^{III}/Co^{III}Co^{III} redox reaction for the RuCoO_x prior to the OER onset potentials. This means that the Ru_{oct}-O-Co_{oct} collaborative coordination with the strong electron coupling effect could promote the electrooxidation behavior of the adjacent Cotet sites but not Cooct and Ruoct sites and thus capture electrons from oxidized Cotet atoms so as to avoid the overoxidation of itself, well in line with the Co 2p XPS result. It should be noted that the failure of identifying the other redox couples from the CV of the RuCoO_x does not necessarily mean that these redox reactions do not proceed. The oxidation currents might simply overlap with the OER currents and thus are difficult to differentiate. The inference that the overoxidation of active Co^{III} and Ru^{IV} sites in the RuCoO_x is significantly suppressed, however, is also supported by the metal dissolution result before and the in situ Raman analysis later. As a result, the RuCoO_x catalyst can maintain a decent acidic OER stability at 10 or even 100 mA cm^{-2} .

The outstanding stability of the RuCoO_x can also be confirmed by the in situ Raman analysis. Besides the emerged Raman signal assigned to dioxygen radical coupling and the peak at 936 cm⁻¹ originating from ClO₄⁻ in the electrolyte, the characteristic Raman peaks of the RuCoO_x do not change or shift as the potential increases (Figure 3f), suggesting the high structural stability of the RuCoO_x under positive bias. In sharp contrast, an extra Raman band at around 580 cm⁻¹ that is raised from the acidic soluble CoO_2 species 50,62,64 is observed at ≥ 1.36 V vs RHE for the Co₃O₄ (Figure 5e). This result is

almost consistent with the CV measurement (Figure 5d) and well explains the different acidic OER stability of the ${\rm Co_3O_4}$ and ${\rm RuCoO_x}$ (Figure 4d). It is noted that the weakening or even disappear of the ${\rm CoO_2}$ characteristic peak at ≥ 1.56 V vs RHE is mainly ascribed to the high solubility of ${\rm CoO_2}$ in acid. In addition, when the OER occurs at higher potentials, bubbles will be accumulated on the electrode surface, which would also reduce the Raman signal. It is also worth noting that the Raman signal corresponding to dioxygen radical coupling also emerges at high potentials (e.g., 1.76 and 1.86 V vs RHE) for the ${\rm Co_3O_4}$. This is not surprising since the ${}^{18}{\rm O}$ isotope-labeled DEMS experiment has previously demonstrated that ${\rm Co_3O_4}$ spinel can realize the diatomic O–O radical coupling in acid. ${}^{31}{\rm O}$

The dissolution of Co species can further be indirectly reflected by the change in the Co^{III}/Co^{II} ratio (corresponding to integral area of A_{1g}/F_{2g} peaks) based on the *in situ* Raman analysis. The Co^{III}/Co^{II} ratio of the $RuCoO_x$ only slightly elevates upon the OER potentials (Figure 5f), suggesting the highly structural integrity of the RuCoO_x under positive bias, well in line with the Co 2p XPS and CV analysis (Figure 5b,d). However, the ratio of Co^{III}/Co^{II} of the Co₃O₄ varies dramatically that it sharply rises to maximum at 1.46 V vs RHE and afterward significantly drops under higher potentials. The former can be attributed to the oxidation behavior of Co^{II}Co^{III}/Co^{III}Co^{III}, while the latter can be ascribed to the overoxidation of Co^{III}Co^{III} to the acidic soluble Co^{IV}Co^{IV} (i.e., CoO₂) that results in a sharp decrease in Co_{oct} content. The difference in the Co^{III}/Co^{II} ratio of the Co_3O_4 and $RuCoO_x$ during the dynamic OER process also confirms the highly structural integrity of the Ru_{oct}-O-Co_{oct} collaborative coordination in the RuCoO_x and thus the high acidic OER stability.

CONCLUSIONS

In summary, we have developed a highly active and robust RuCoO_x catalyst through partial substitution of Co_{oct} sites in spinel Co₃O₄ by Ru atoms for enhanced acidic OER performance. Both the experimental and theoretical results confirmed that the resulted Ru_{oct}-O-Co_{oct} collaborative coordination with strong electron coupling effect can proceed the thermodynamically breakthrough heterogeneous DOM pathway. Meanwhile, the Ruoct-O-Cooct collaborative coordination can also gain electrons from surrounding electronenriched Cotet sites via bridging oxygen atoms, so that the overoxidation of active Ruoct and Cooct centers into acidic soluble RuO₄ and CoO₂ species can be suppressed. As a result, the RuCoO_x catalyst with an ultralow Ru loading of 42.8 μ g cm⁻² delivers decent OER performance with only a 200 mV overpotential and a low degradation rate of 0.45 mV h⁻¹ at 10 mA cm⁻². Our work not only establishes the RuCoO_x as an efficient electrocatalysts for acidic OER, but also suggests an effective strategy to dramatically boost the acidic OER performance of Co₃O₄ and potentially the big family of spinel oxides by site preferential atomic substitution.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/jacs.3c05556.

Experimental section, theoretical calculation, and additional figures and tables (PDF)

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Notes

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ABBREVIATIONS

OER oxygen evolution reaction

PEMWE proton exchange membrane water electro-

lyzer

LOM lattice oxygen mechanism
AEM adsorbate evolution mechanism
DOM diatomic oxygen mechanism

Co_{oct} octahedral Co Co_{tet} tetrahedral Co

SEM scanning electron microscopy SAED selected area electron diffraction

HAADF-STEM high-angle annular dark-field scanning trans-

mission electron microscopy

XRD X-ray diffraction

DFT density functional theory
XAS X-ray absorption spectrometry
EPR electron paramagnetic resonance

CV cyclic voltammetry
LSV linear sweep voltammetry

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