

# Achieving Efficient CO<sub>2</sub> Electrolysis to CO by Local Coordination Manipulation of Nickel Single-Atom Catalysts

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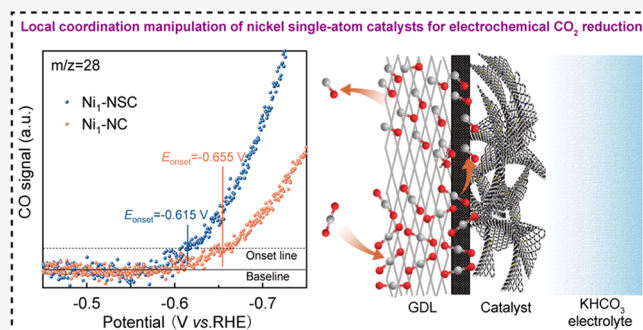
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Supporting Information

**ABSTRACT:** Selective electroreduction of CO<sub>2</sub> to C<sub>1</sub> feed gas provides an attractive avenue to store intermittent renewable energy. However, most of the CO<sub>2</sub>-to-CO catalysts are designed from the perspective of structural reconstruction, and it is challenging to precisely design a meaningful confining micro-environment for active sites on the support. Herein, we report a local sulfur doping method to precisely tune the electronic structure of an isolated asymmetric nickel–nitrogen–sulfur motif (Ni<sub>1</sub>-NSC). Our Ni<sub>1</sub>-NSC catalyst presents >99% faradaic efficiency for CO<sub>2</sub>-to-CO under a high current density of −320 mA cm<sup>−2</sup>. *In situ* attenuated total reflection surface-enhanced infrared absorption spectroscopy and differential electrochemical mass spectrometry indicated that the asymmetric sites show a significantly weaker binding strength of \*CO and a lower kinetic overpotential for CO<sub>2</sub>-to-CO. Further theoretical analysis revealed that the enhanced CO<sub>2</sub> reduction reaction performance of Ni<sub>1</sub>-NSC was mainly due to the effectively decreased intermediate activation energy.

**KEYWORDS:** Electrochemical CO<sub>2</sub> reduction, asymmetric coordination, sulfur, isolated nickel atom



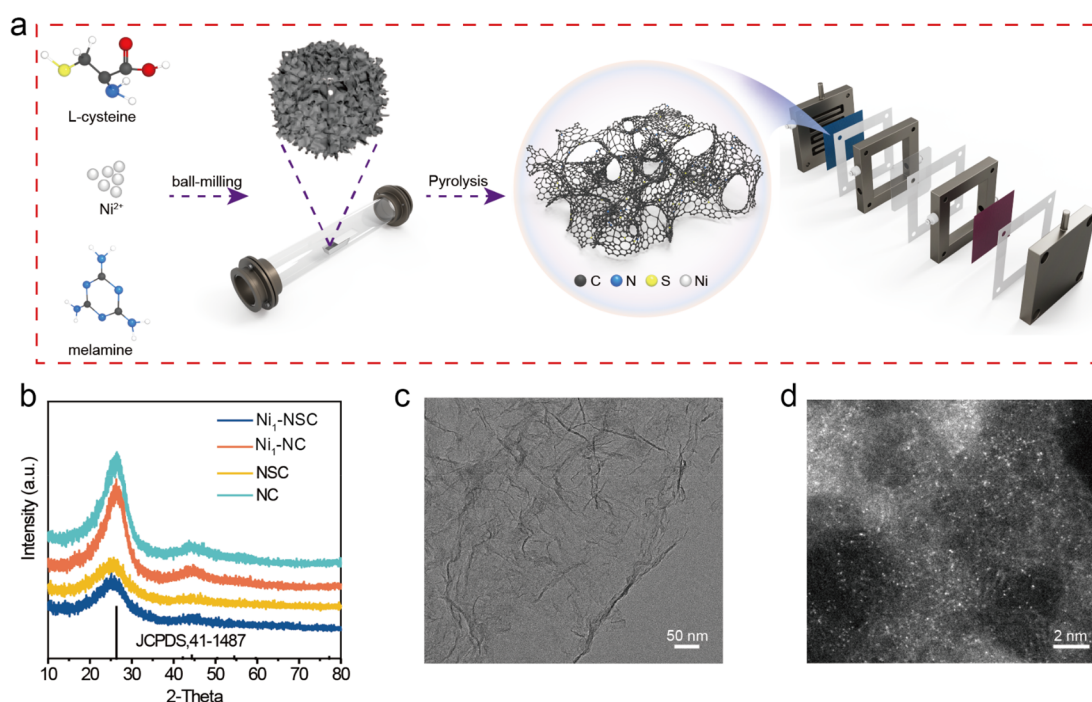
The demanding use of fossil fuels in modern industrial society has produced excessive emissions of CO<sub>2</sub> and caused serious climate-related problems.<sup>1–5</sup> It is highly desirable to develop CO<sub>2</sub> utilization technologies, such as electrochemical CO<sub>2</sub> reduction (ECR). ECR is powered by renewable electricity and has been considered to be a promising CO<sub>2</sub> recycling technology that can efficiently convert CO<sub>2</sub> into value-added compounds.<sup>6–9</sup> Among various products from ECR, carbon monoxide (CO) is one of the most valuable products due to its ease of separation and high economic value per electron, and CO can be used as a fundamental feedstock for the production of many essential chemicals via the Fischer–Tropsch chemistry, or as a reducing agent in industrial metallurgy.<sup>10,11</sup> However, operating current catalysts for CO<sub>2</sub>-to-CO at a commercially relevant scale remains challenging. Thus, designing a durable catalyst that has a high CO selectivity at large current densities is of paramount importance. In a general CO<sub>2</sub>-to-CO process, the carbon atoms of CO<sub>2</sub> molecules (instead of oxygen atoms) are first adsorbed on the catalyst surface and form the \*COOH intermediate via a proton-coupled electron transfer process; then, the adsorbed \*COOH is further reduced to \*CO and ultimately desorbed from the electrode.<sup>12–14</sup> Therefore, ideal CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) catalysts for CO evolution should not only have a unique electronic structure showing a decreased barrier for \*COOH formation and proper binding

strength with the formed \*COOH but also bind weakly enough with the produced \*CO on the catalyst surface.

To date, great efforts have been made in the search for active and durable CO-efficient catalysts. Nanostructured Ag and Au are state-of-the-art catalysts for the CO<sub>2</sub>-to-CO process, but aside from their high activities, their cost and low faradaic efficiency at high current densities limited their wide industrial applications.<sup>15–18</sup> Recently, earth-abundant transition-metal (TM) catalysts, such as Fe, Co, Ni, and Cu, come out as efficient and cheaper catalysts for CO<sub>2</sub>-to-CO, but their activity and selectivity are far from satisfactory.<sup>19–21</sup> To address this problem, downsizing TM into individual atoms and tuning their valence states via well-defined supports have been demonstrated to be effective. Specifically, TM single atoms (TM-SAs) anchored on nitrogen-doped carbon carriers are considered to be one of the most promising nonprecious metal catalysts to convert CO<sub>2</sub> into CO.<sup>22–24</sup> The plane-symmetric M–N<sub>4</sub> structure, where the metal center is coordinatively bonded to four N atoms, was shown to act as the active site of

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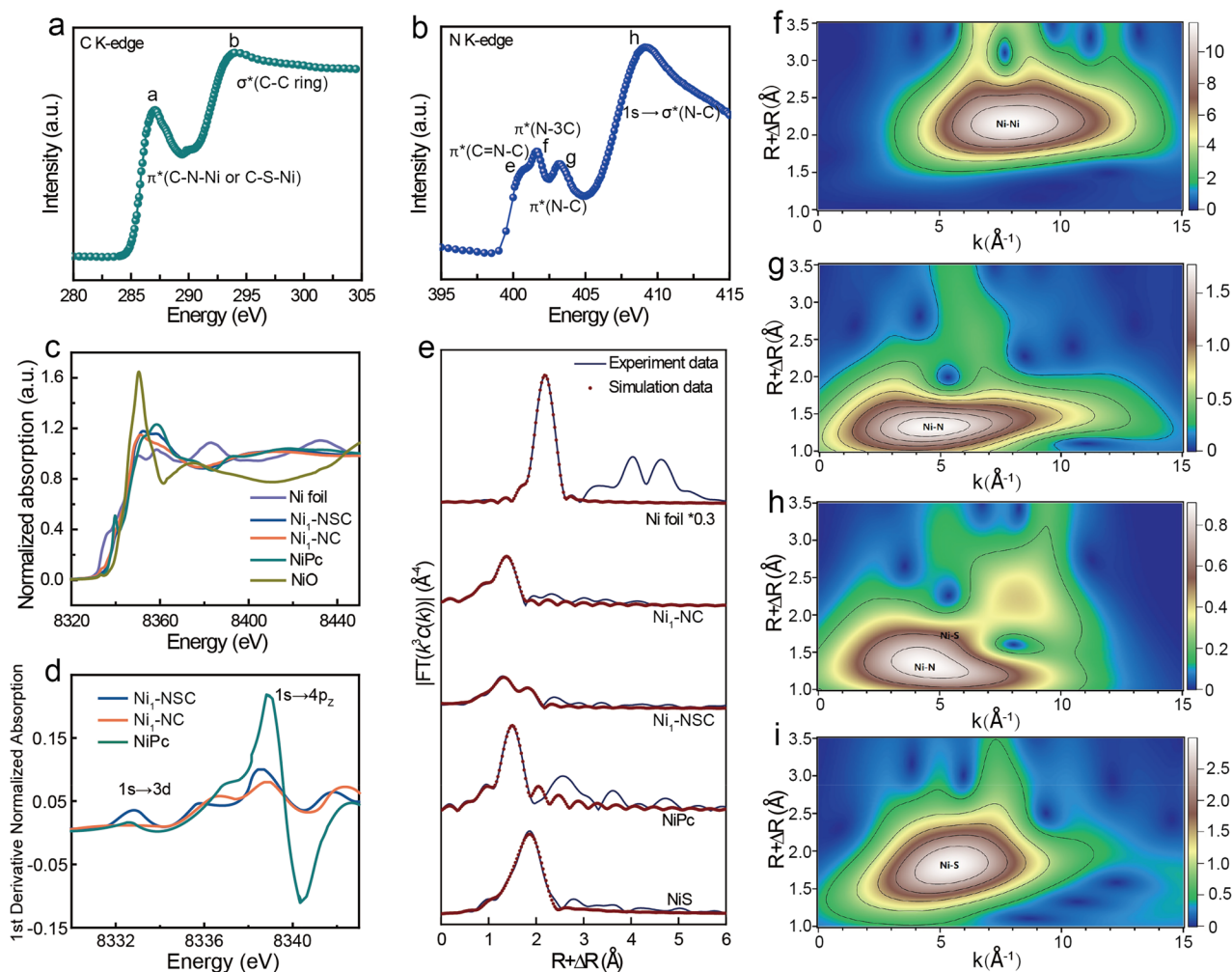
**Figure 1. Preparation and structural characterization of Ni<sub>1</sub>-NSC.** (a) Schematic illustration displaying the preparation process of Ni<sub>1</sub>-NSC and application in CO<sub>2</sub>RR. (b) X-ray diffraction patterns. (c) TEM image of Ni<sub>1</sub>-NSC. (d) HAADF-STEM image of Ni<sub>1</sub>-NSC.

CO<sub>2</sub>RR and have the advantage to suppress hydrogen evolution reaction (HER).<sup>25,26</sup> However, a symmetrical planar coordinated structure exhibits a large symmetric electron distribution and is not conducive to axial CO<sub>2</sub> adsorption and activation, resulting in high kinetic overpotentials for CO<sub>2</sub>-to-CO. To break the symmetry, heteroatoms that have weaker electronegativity toward N can be doped, which can coordinate directly with metal centers and tune the catalytic activity by adjusting the electron-withdrawing/donating properties.<sup>27</sup> The latest research shows that sulfur is a good heteroatom candidate, since sulfur doping is more conducive to promoting the dissociation of water and offers enough protons to convert CO<sub>2</sub> into \*COOH.<sup>28</sup> Herein, we employed a ball-milling strategy to create atomically dispersed nickel within a carbon matrix co-doped with nitrogen and sulfur. X-ray absorption spectroscopy (XAS) analysis indicated that the catalyst mainly contained asymmetrically coordinated Ni<sub>1</sub>-N<sub>3</sub>-S motifs, where the microenvironment of the metal atom was regulated by directly coordinating with the S atom. This novel catalyst shows superior activity and efficiency (nearly 100% CO selectivity at -320 mA cm<sup>-2</sup> in a flow cell) over the traditional Ni<sub>1</sub>-N<sub>4</sub> moiety. *In situ* attenuated total reflection surface-enhanced infrared absorption spectroscopy (ATR-SEIRAS) and differential electrochemical mass spectrometry (DEMS) suggest that the asymmetrically coordinated Ni<sub>1</sub>-N<sub>3</sub>-S sites show a lower kinetic overpotential for CO<sub>2</sub>-to-CO and ensure improved CO yield at high current densities. Density functional theory (DFT) calculations further reveal that the enhanced activity and selectivity are attributed to the optimal Gibbs free energy of the \*COOH intermediate on the Ni<sub>1</sub>-N<sub>3</sub>-S sites. Overall, this work demonstrates the significant selectivity and activity advantage of the Ni<sub>1</sub>-NSC catalyst and provides unique insights into the design of other asymmetrically coordinated active sites for CO<sub>2</sub>RR.

We synthesized the Ni<sub>1</sub>-NSC catalyst by grinding a mixture of L-cysteine, melamine, and nickel nitrate into a homogeneous

precursor, followed by a two-step high-temperature calcination under an Ar atmosphere (Figure 1a). For comparison, Ni<sub>1</sub>-NC was prepared in the same procedure, whereas L-cysteine was replaced by L-alanine. The corresponding catalysts without metals are denoted as NSC and NC. We first investigated the crystallinity and crystalline phase of the catalyst by using X-ray diffraction (XRD), and these four catalysts display similar main diffraction peaks at 26.2° and 43.6°, which were assigned to the (002) and (101) planes of graphitic carbon (Figure 1b). No diffraction peaks for residual Ni, NiN<sub>x</sub>, or NiC<sub>x</sub> were detected in either the Ni<sub>1</sub>-NSC or Ni<sub>1</sub>-NC samples. On the other hand, Raman spectroscopy shows a higher I<sub>D</sub>/I<sub>G</sub> ratio of Ni<sub>1</sub>-NSC (1.002) compared with Ni<sub>1</sub>-NC (0.964), suggesting its more disordered carbon structure (Figure S1). The typical morphology of Ni<sub>1</sub>-NSC exhibits thin layered structures, and no particles or clusters were observed (Figure 1c and Figure S2a,b). Energy dispersive spectroscopy (EDS) revealed that C, N, S, and Ni were distributed uniformly across the entire carbon support (Figure S2c,d). The mass ratio of Ni within different samples was measured using inductively coupled plasma optical emission spectroscopy (ICP-OES), showing 0.486 and 0.424 wt % for Ni<sub>1</sub>-NSC and Ni<sub>1</sub>-NC, respectively. The nanostructure of Ni<sub>1</sub>-NSC was further revealed by aberration-corrected high-angle annular dark field scanning transmission electron microscopy (HAADF-STEM), and monodispersed Ni atoms were confirmed by the isolated bright dots across the entire substrate (Figure 1d). Additional morphology and mapping analyses of the four catalysts are shown in Figures S2–S5.

We further performed X-ray photoelectron spectroscopy (XPS) to analyze the chemical states and bonding configurations of the as-synthesized catalysts. The Ni 2p spectrum of the Ni<sub>1</sub>-NSC sample shows a positive shift of the Ni 2p<sub>3/2</sub> main peak (854.9 eV) relative to Ni metal (852.6 eV), indicating the positive oxidation state of single Ni atoms (Figure S6a).<sup>29</sup> The C 1s spectrum displays an asymmetric shape that can be

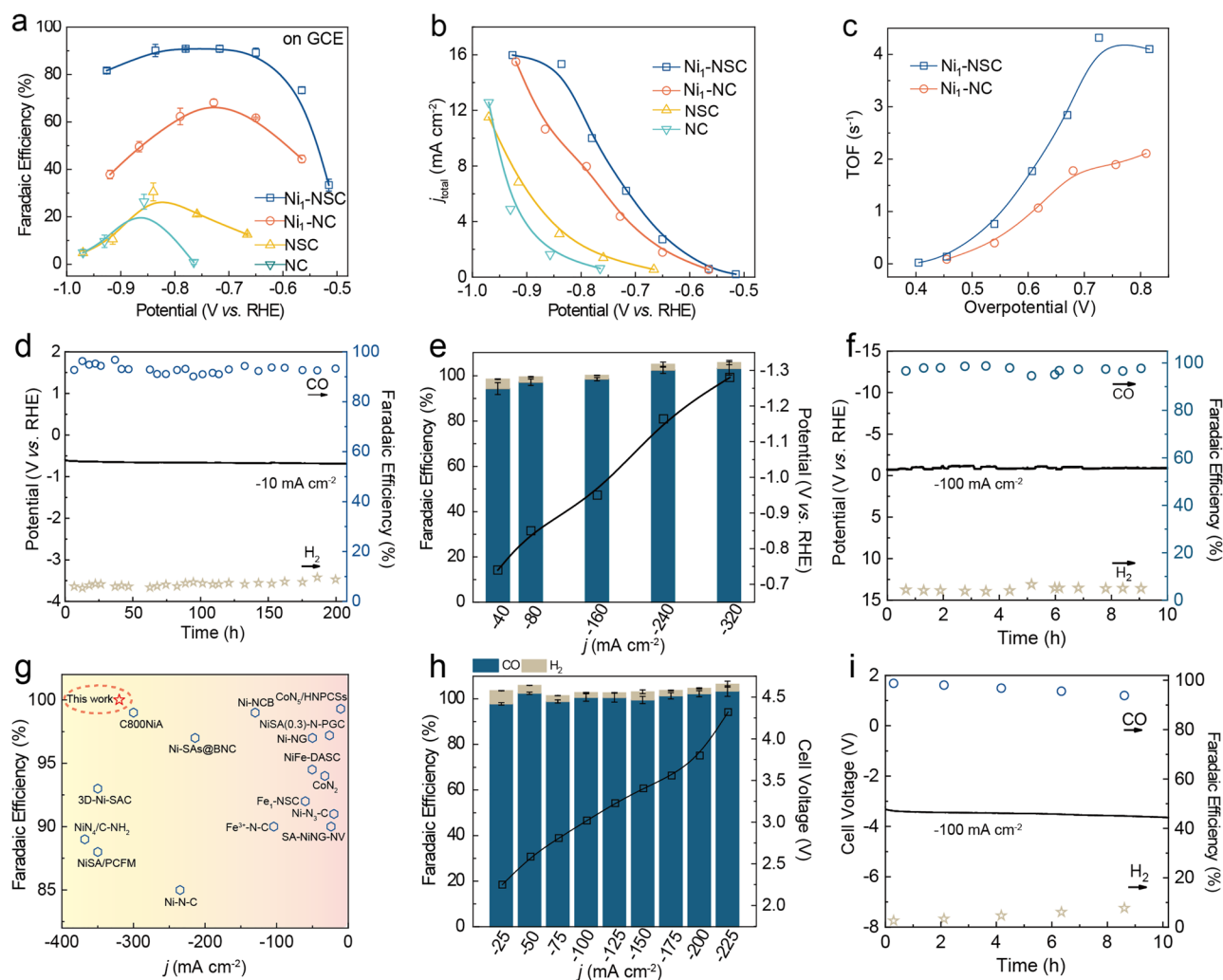


**Figure 2. Structural characterization of catalysts.** (a) C K-edge XANES spectra for NiI-NSC and (b) N K-edge XANES spectra for NiI-NSC. (c) The experimental Ni K-edge XANES spectra of NiI-NSC, NiI-NC, and the references (Ni foil, NiPc, and NiO). (d) First derivative of the pre-edge region of spectra in NiI-NSC, NiI-NC, and NiPc. (e) Fourier transformed EXAFS spectra in the R space and the corresponding FT-EXAFS fitting curves. (f–i) WT-EXAFS plots of Ni foil, NiI-NC, NiI-NSC, and NiS.

separated into two C species, including C=C (284.6 eV) and C–C/C=N (285.4 eV) (Figure S6b). The high-resolution S 2p XPS spectrum shows two main peaks centered at 163.7 and 164.9 eV, which were assigned to S 2p<sub>3/2</sub> and S 2p<sub>1/2</sub>, respectively (Figure S6c).<sup>30</sup> Afterward, soft X-ray absorption near-edge structure (XANES) spectroscopy was carried out for NiI-NSC. The C K-edge XANES spectrum (Figure 2a) shows excitation resonances at 287.2 and 293.6 eV, which could be assigned to the dipole transition of the C 1s core electron to the  $\pi^*$  (C–N–Ni) or  $\pi^*$  (C–S–Ni) and  $\sigma^*$  (C–C ring) orbitals.<sup>31</sup> Figure 2b shows the N K-edge spectrum and the peak at the lower energy side, and the signals are assigned to the  $1s \rightarrow \pi^*$  transition of heterocyclic rings ( $\pi^*$ (C=N–C)), graphitic 3-fold nitrogen atoms ( $\pi^*$ (N–3C)), sp<sup>3</sup> N–C bridging among tri-s-triazine moieties ( $\pi^*$ (N–C)), and the  $1s \rightarrow \sigma^*$ (N–C) transition, respectively.<sup>32,33</sup> The detailed N 1s spectra in NiI-NSC (Figure S6d) can be separated into four N species, including graphitic N (401.3 eV), pyrrolic N (400.6 eV), Ni–N (399.2 eV), and pyridinic N (398.1 eV).<sup>34</sup> Similar results were also obtained for the NiI-NC catalyst (Figure S7). Furthermore, X-ray absorption fine structure (XAFS) measurements were carried out to probe the chemical identity of the Ni atoms. In Figure 2c, the absorption edges of NiI-NSC and NiI-

NC are located between Ni foil and NiO, indicating that the Ni atoms possess positive charges and that the average oxidation state is between the two references. The electronic structures of the catalysts were further studied via the first derivative XANES spectra (Figure 2d). The  $1s \rightarrow 3d$  transition peaks at 8334 eV are evident on NiI-NSC, and the peak intensity is significantly stronger in NiI-NSC compared to nickel phthalocyanine (NiPc) or NiI-NC. The increased peak intensity in NiI-NSC indicates that the Ni microenvironment is not uniformly centrosymmetric and is deviating from the perfect plane-symmetric morphology.<sup>35,36</sup> The Fourier transform (FT) EXAFS spectra of both NiI-NSC and NiI-NC displayed the main peak at 1.4 Å, which was attributed to the scattering of the Ni–N coordination shell. Notably, no related peak corresponding to the Ni–Ni bond (2.2 Å) was observed in either NiI-NSC or NiI-NC (Figure 2e and Figure S8). Surprisingly, a shoulder peak located at 1.86 Å was also detected in NiI-NSC, and the peak was considered to be derived from the Ni–S scattering. Wavelet transform (WT) plots were further applied to identify the Ni K-edge EXAFS for a better visual comparison of the catalysts (Figure 2f–i and Figure S9). It was found that Ni foil displays a prominent Ni–Ni bond at approximately 8.0 Å<sup>−1</sup>, while NiI-NSC and NiI-NC



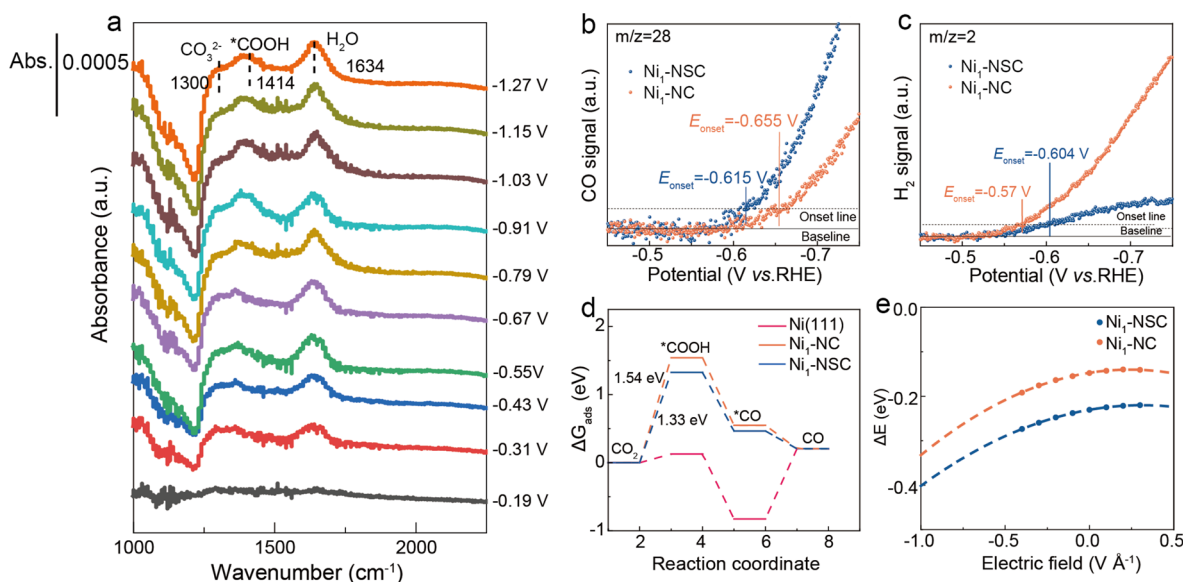


**Figure 3.** CO<sub>2</sub> electroreduction performance. (a) FEs of CO under different applied potentials for Ni<sub>1</sub>-NSC, Ni<sub>1</sub>-NC, NSC, and NC. (b) The corresponding  $j$ - $V$  curve. (c) TOF of Ni<sub>1</sub>-NSC and Ni<sub>1</sub>-NC in an H-cell test based on the total mass loaded on the electrode. (d) The CO<sub>2</sub>RR stability test of Ni<sub>1</sub>-NSC in an H-cell at  $-10 \text{ mA cm}^{-2}$ . (e) The CO<sub>2</sub> performance of Ni<sub>1</sub>-NSC on a gas diffusion layer electrode in a flow cell configuration. (f) Stability test at  $-100 \text{ mA cm}^{-2}$  current density of Ni<sub>1</sub>-NSC. (g) Contrasting the FEs and current density values for Ni<sub>1</sub>-NSC and other reported single-atom catalysts (Table S2). (h) The corresponding FEs of H<sub>2</sub> and CO in an anion membrane electrode assembly and the corresponding  $j$ - $V$  curve without  $iR$ -compensation of Ni<sub>1</sub>-NSC. (i) Long-term stability test at  $-100 \text{ mA cm}^{-2}$  current density in MEA of Ni<sub>1</sub>-NSC.

feature an intensity maximum at  $4.5 \text{ \AA}^{-1}$ , which closely matches the Ni—N bonding. This further indicates that Ni of Ni<sub>1</sub>-NSC and Ni<sub>1</sub>-NC is atomically dispersed without aggregation. From the WT contour plots of Ni<sub>1</sub>-NSC and NiS standards, the WT signals derived from the Ni—S bond are located at  $\sim 5.5 \text{ \AA}^{-1}$ . Quantitative EXAFS fitting curves (Figure 2e and Table S1) indicate that Ni in Ni<sub>1</sub>-NSC is coordinated with three N atoms and one S atom with unsymmetrical coordination (Ni<sub>1</sub>-N<sub>3</sub>-S). For the Ni<sub>1</sub>-NC catalyst, four N atom coordination structures were observed.

The CO<sub>2</sub>RR performance of the catalysts was first investigated in a three-electrode H-type electrochemical cell using a  $0.5 \text{ M KHCO}_3$  electrolyte. The catalyst was immobilized on a mirror-polished glassy-carbon electrode (GCE). Online gas chromatography was used to quantify the faradaic efficiency (FE) of gaseous products, and CO was demonstrated to be the major detectable product, while the residual faradaic charge was consumed in the HER process. Notably, the Ni<sub>1</sub>-NSC catalyst showed a good CO FE of 73% at a low potential of  $-0.57 \text{ V}$  versus a reversible hydrogen electrode (vs RHE). As the potential becomes more negative,

the selectivity of CO increases, and that of H<sub>2</sub> decreases correspondingly. The FEs of CO were more than 90% over a potential range of  $-0.65$  to  $-0.82 \text{ V}$  vs RHE. Specifically, Ni<sub>1</sub>-NSC displays a maximum FE<sub>CO</sub> of 92% at  $-0.78 \text{ V}$  vs RHE, far surpassing Ni<sub>1</sub>-NC (69% at  $-0.73 \text{ V}$ ), NSC (32% at  $-0.84 \text{ V}$ ), and NC (27% at  $-0.86 \text{ V}$ ) (Figure 3a and Figure S10). The results clearly indicated that the Ni<sub>1</sub>-N<sub>3</sub>-S moieties play a dominating role in the catalytic process. Ni<sub>1</sub>-NSC also exhibits a higher overall current density or partial current density of CO (Figure 3b and Figure S11) than that of the studied catalysts. To further evaluate the intrinsic activities of Ni<sub>1</sub>-NSC and Ni<sub>1</sub>-NC in the CO<sub>2</sub>RR, the CO production turnover frequency (TOF) is calculated in accordance with the total mass of the loaded Ni on the electrode (see details in the Experimental Section). As illustrated in Figure 3c, the Ni<sub>1</sub>-NSC shows much higher TOF values relative to Ni<sub>1</sub>-NC. To examine the stability of Ni<sub>1</sub>-NSC, we performed a stationary long-term continuous electrocatalytic test at  $-10 \text{ mA cm}^{-2}$ . After continuous operation for approximately 210 h, the potential and FE<sub>CO</sub> of the Ni<sub>1</sub>-NSC catalyst were almost



**Figure 4.** Mechanistic studies of electrochemical  $\text{CO}_2$ -to- $\text{CO}$  conversion. (a) *In situ* ATR-SEIRAS spectra recorded at different applied potentials for  $\text{Ni}_1\text{-NSC}$ , Abs., absorbance. All potentials were calibrated to the RHE scale. (b, c) *In situ* DEMS measurement of  $\text{CO}$  and  $\text{H}_2$  production during the  $\text{CO}_2\text{RR}$  on  $\text{Ni}_1\text{-NSC}$  and  $\text{Ni}_1\text{-NC}$  catalysts. (d) Free-energy diagram for different Ni-centered moieties. (e) The effect of electric field on  $\text{*CO}$  desorption during electrocatalytic  $\text{CO}_2$  reduction of  $\text{Ni}_1\text{-NSC}$  and  $\text{Ni}_1\text{-NC}$  catalysts.

unchanged, suggesting that it possesses excellent intrinsic stability (Figure 3d).

In H-type cells, due to the solubility limit of  $\text{CO}_2$ , the concomitant HER side reactions are unavoidable. Moreover, the large interelectrode distance and small electrode surface area cannot meet the industrial requirements, making it hard to achieve industrial current densities.<sup>37</sup> To assess the performance of the catalysts for industrial ECR, a flow cell equipped with a gas diffusion electrode (GDE) was applied, where the porous nature of the GDE provides enough three-phase interfaces between the gaseous  $\text{CO}_2$  and catalysts.<sup>38</sup> As shown in Figure 3e,  $\text{Ni}_1\text{-NSC}$  achieved a high current density of  $-320 \text{ mA cm}^{-2}$  at  $-1.28 \text{ V}$  vs RHE, and nearly 100%  $\text{CO}$  selectivity can still be maintained at such high current densities. More importantly, the voltage and  $\text{FE}_{\text{CO}}$  remained unchanged after continuous electrolysis for 10 h at  $-100 \text{ mA cm}^{-2}$  (Figure 3f). We further compared the  $\text{CO}_2\text{RR}$  activity of  $\text{Ni}_1\text{-NSC}$  with other reported transition metal single-atom catalysts and found that  $\text{Ni}_1\text{-NSC}$  exhibited superior activities (Figure 3g, Figure S12a–c, and Table S2). An anion membrane electrode assembly (MEA) device was also used to evaluate the activity of the catalysts. As shown in Figure 3h, significant  $\text{CO}_2\text{RR}$  occurs above the 2.25 V cell voltage and quickly reaches a high current density of  $-225 \text{ mA cm}^{-2}$  at 4.32 V without  $iR$ -compensation. It was worth noting that the competitive HER reaction is greatly suppressed (2%  $\text{H}_2$ ) due to the fast  $\text{CO}_2$  diffusion, and a high  $\text{CO}$  selectivity ( $\sim 99\%$ ) was achieved over a wide range of potentials toward other samples ( $\text{Ni}_1\text{-NC}$ ,  $\text{NSC}$ , and  $\text{NC}$ ) (Figure S12d–f). The long-term stability test of  $\text{Ni}_1\text{-NSC}$  manifested that the FE for  $\text{CO}$  was maintained around 94% after continuous electrolysis at  $-100 \text{ mA cm}^{-2}$  using a MEA device. The slight potential fluctuation was probably due to the membrane degradation or salt precipitation. Additionally, we have also characterized the post-catalysis  $\text{Ni}_1\text{-NSC}$  (Figures S13 and S14). After the stability test, the morphology of the catalyst and its atomic valence state did not change significantly. Collectively, these findings serve as robust evidence affirming its superior stability.

To identify the catalytic mechanism and determine the absorbed surface intermediates with the catalyst, we performed *in situ* ATR-SEIRAS at different potentials (Figure 4a). An obvious broad and asymmetric peak at  $\sim 1414 \text{ cm}^{-1}$  was detected, which can be assigned to  $\text{*COOH}$ .<sup>39</sup> As expected, the  $\text{*COOH}$  peak gradually appeared and became more apparent as the reduction potential became more negative, proving the direct uptake of  $\text{CO}_2$  molecules from the electrolyte and their reduction through the proton-coupled electron transfer process. On the contrary, we did not observe the signal of  $\text{*CO}$  intermediates on the surface. This could be attributed to the weakened adsorption of  $\text{*CO}$  intermediates on the  $\text{Ni}_1\text{-NSC}$  surface.<sup>40</sup> *In situ* DEMS was further applied to directly detect the electrochemical formation of  $\text{CO}$ . As shown in Figure 4b,c, we can clearly see that the introduction of sulfur can significantly lower the kinetic overpotentials and promote  $\text{CO}$  generation while suppressing the production of  $\text{H}_2$ . This result is consistent with the results of the  $\text{CO}_2\text{RR}$  test. To elucidate the reaction mechanism of the enhanced  $\text{CO}_2\text{RR}$  activity of  $\text{Ni}_1\text{-NSC}$ , the detailed reaction free-energy diagrams of  $\text{Ni}_1\text{-NSC}$ ,  $\text{Ni}_1\text{-NC}$ , and  $\text{Ni}(111)$  are presented in Figure 4d. Two proton–electron transfer steps are included in electrochemical  $\text{CO}_2$ -to- $\text{CO}$  conversion. Finally, the formed  $\text{*CO}$  is desorbed from the catalyst surface. Figure S15 shows the possible  $\text{CO}_2\text{RR}$  process of the  $\text{Ni}_1\text{-NSC}$  catalyst. The first protonation step of  $\text{CO}_2$  to form  $\text{*COOH}$  is usually considered to be the rate-determining step. The free-energy change ( $\Delta G$ ) for  $\text{CO}_2$ -to- $\text{*COOH}$  on  $\text{Ni}_1\text{-NSC}$  (1.33 eV) is significantly smaller than that on  $\text{Ni}_1\text{-NC}$  (1.54 eV), proving that the asymmetric coordination structure is conducive to lower the reaction energy barrier. Simultaneously, the free-energy diagrams on Ni nanoparticles (111) for the  $\text{CO}_2\text{RR}$  to  $\text{CO}$  are also calculated. Although its energy to form  $\text{*COOH}$  is lower, the desorption energy of  $\text{*CO}$  is too strong (1.17 eV). Furthermore, we incorporated additional details regarding the impact of the electric field on  $\text{*CO}$  desorption during the electrocatalytic  $\text{CO}_2$  reduction process over  $\text{Ni}_1\text{-NSC}$  and  $\text{Ni}_1\text{-NC}$  catalysts (Figure 4e). From the calculation results, it can

be seen that the intrinsic dipole moment and polarizability of the adsorbents on the two catalysts ( $\text{Ni}_1\text{-NSC}$  and  $\text{Ni}_1\text{-NC}$ ) are similar. Therefore, the localized sulfur ligand does not significantly influence the adsorbent's susceptibility to the electric field during the  $\text{CO}_2\text{RR}$ . It means that the electric field has little effect on  $^*\text{CO}$  desorption.

In summary, we have developed a facile way to produce atomically dispersed Ni on N, S co-doped carbon carriers. The catalyst outperformed other Ni-based single-atom catalysts with >99%  $\text{CO}$  faradaic efficiency and  $\text{CO}$  partial current densities of  $>-320\text{ mA cm}^{-2}$ , and an MEA device presents >99%  $\text{FE}_{\text{CO}}$  in a high current density of  $-225\text{ mA cm}^{-2}$ , underlining a promising future for industrial applications. The critical role of the asymmetric  $\text{Ni}_1\text{-N}_3\text{-S}$  sites was demonstrated by the dramatically decreased activity for the  $\text{Ni}_1\text{-N}_4$  moiety. *In situ* ATR-SEIRAS and DEMS measurements indicated that the introduction of sulfur can significantly lower the kinetic overpotential and promote faster  $\text{CO}$  desorption. Further theoretical analysis revealed that the enhanced  $\text{CO}_2\text{RR}$  performance of  $\text{Ni}_1\text{-NSC}$  was mainly due to the effectively decreased intermediate activation energy. This work not only offers new insights into the role of coordination environment regulation in governing the  $\text{CO}_2\text{RR}$  but also provides a reliable design rule for single-atom catalysts.

## ■ ASSOCIATED CONTENT

### SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.nanolett.3c01808>.

Figures S1–S15 describing the details of the experimental and theoretical sections, Raman spectra, TEM, STEM images, EDS mappings, HAADF-STEM images, XPS, XANES, electrochemical data, and the reaction mechanism and Tables S1 and S2 describing the EXAFS fitting parameters and a comparison of the  $\text{CO}_2\text{RR}$  performance (PDF)

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### Notes

The authors declare no competing financial interest.

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